

1 Effective Wetting Area Based on Electrochemical
2 Impedance Analysis: Hydrophilic Structured Surface

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1 ABSTRACT: Wettability on nano/microstructured surfaces is gaining remarkable interest for a
2 wide range of applications; however, little is known about the effective wetting area of the solid–
3 liquid interface. In this study, the effect of wettability on electrochemical impedance was
4 experimentally investigated to obtain a better understanding of the effective wetting area. We
5 demonstrate that the water contact angle decreases significantly at hydrophilic surfaces with denser
6 nano/microstructures. Based on the analysis of equivalent electrical circuits, we found that the
7 electrochemical impedance decreases with reducing the water contact angle, showing a
8 dependence on the effective wetting area, i.e., the real solid–liquid contact area. Also, the charge
9 transfer resistance at low frequency was found to be the dominant parameter to estimate the
10 effective wetting area at the solid–liquid interface.

11 INTRODUCTION

12 The wettability of solid surfaces has attracted considerable interest since a long period.¹⁻⁶ A large
13 number of special wettability materials is being widely used in academic science and industrial
14 technologies.⁷⁻¹² Recent advances in microfluidic devices¹³⁻¹⁶ and biochips¹⁷⁻¹⁹ have facilitated the
15 fabrication of hydrophilicity and super-hydrophilicity at the solid–liquid interface, where the
16 hydrophilicity and super-hydrophilicity are defined when the water contact angles (WCAs) are
17 smaller than 90° and 5°, respectively.²⁰

18 Physical morphology and chemical composition are important factors that control surface
19 wettability.²¹⁻²⁵ For a static droplet on structured surfaces, Wenzel model describes the state in
20 which the interiors of the surface structures are fully wetted.²⁶ Conversely, the Cassie–Baxter
21 model describes a completely non-wetted state.²⁷ An intermediate wetting state between the two
22 classical wetting models, i.e., partial wetting state, has been proposed by Nagayama et al. based

1 on the molecular dynamics simulation.²⁸ The effective wetting ratio in the interior of
2 nano/microstructures is applied to the partial wetting model; this ratio is 1 for the fully wetted state
3 and 0 for the non-wetted state. However, in many cases, the experimental water contact angles
4 disagree with either theoretical Wenzel or Cassie–Baxter models^{29–31} but correspond well with the
5 partial wetting model for the nano/microstructured surfaces.^{32,33} The dependence of the static
6 contact angle on the effective wetting ratio provides evidence that the solid–liquid contact area of
7 the partial wetting state is larger than that of non-wetted state but smaller than fully wetted state.
8 In order to clarify the effective wetting ratio, the $\Phi_s = \Phi + (r_w - \Phi)f$ was applied to show the area ratio
9 of solid–liquid interface to the apparent flat surface, where Φ is the solid fraction, r_w is the surface
10 area increment ratio and f is the effective wetting ratio.³³ Φ_s result in the surface area increment
11 ratio $r_w (= \Phi + (r_w - \Phi) \times 1 (r_w \geq 1))$ for the fully wetted state; solid fraction $\Phi (= \Phi + (r_w - \Phi) \times 0, (\Phi \leq$
12 1)) for the non-wetted state. Compared with the one-scale structured surface, the solid–liquid
13 contact area at the hierarchical structured surface change from $\Phi_s \times A$ to $\Phi_s \times \Phi_s \times A$ due to the
14 similarity between the primary and secondary structures.³³ Thus, the solid–liquid contact area at
15 the hierarchical structured surface increases from $r_w \times A$ to $r_w \times r_w \times A$ at the fully wetted state but
16 decreases from $\Phi \times A$ to $\Phi \times \Phi \times A$ at the non-wetted state, and depends on the effective wetting ratio
17 f at the partial wetted state. However, the determination of wetting state and effective wetting ratio
18 at the structured surface is still not clear, and the study of effective wetting area is open for
19 questions.

20 Electrochemical impedance spectroscopy (EIS) is a practical technique applied in numerous
21 fields, especially in surface science.^{34–38} EIS provides useful solid–liquid interfacial information
22 based on the signal response under sinusoidal potential excitation at different frequencies.^{39–43} In
23 this work, the effective wetting area at structured surface was studied. Nano/microstructures were

1 prepared on aluminum (Al) surfaces and the static contact angles were measured experimentally.
2 The electrochemical impedance at hydrophilic Al surfaces was evaluated to investigate the solid–
3 liquid contact area. Moreover, the theoretical wetting models were first applied to the equivalent
4 electrical circuits for the electrochemical impedance analysis to estimate the effective wetting area
5 at the structured surface.

6 EXPERIMENTAL SECTION

7 **Sample Preparation.** Chemical etching was used to prepare hydrophilic samples with
8 nano/microstructured surfaces. Al substrates (3003 Al alloy, 20 mm × 15 mm × 0.5 mm) that were
9 previously polished using sandpaper were ultrasonically cleaned with acetone and ethanol
10 sequentially for 10 min, and then rinsed in the deionized water (DI). The substrates were then
11 immersed in 1.25 M NaOH for 10 s to remove the aluminum oxide film from the surface. The
12 etching process was performed by immersing the cleaned substrates in a 1 M CuCl₂ solution for 1,
13 2, 5, 10, or 20 s at room temperature, where they underwent the reactions shown in Eqs. (1) and
14 (2). These samples were designated (b)–(f), respectively, with an unetched control sample being
15 designated (a).



18 Here, Cu²⁺ reacts with Al via a replacement reaction, resulting in a thin layer of Cu being deposited
19 on the Al surface. The thin layer of Cu deposition was clearly eliminated by the ultrasonic bath of
20 the DI water, and only Al was observed on the surface (Figure S1) using X-ray diffraction (D8
21 ADVANCE, Bruker, Germany).

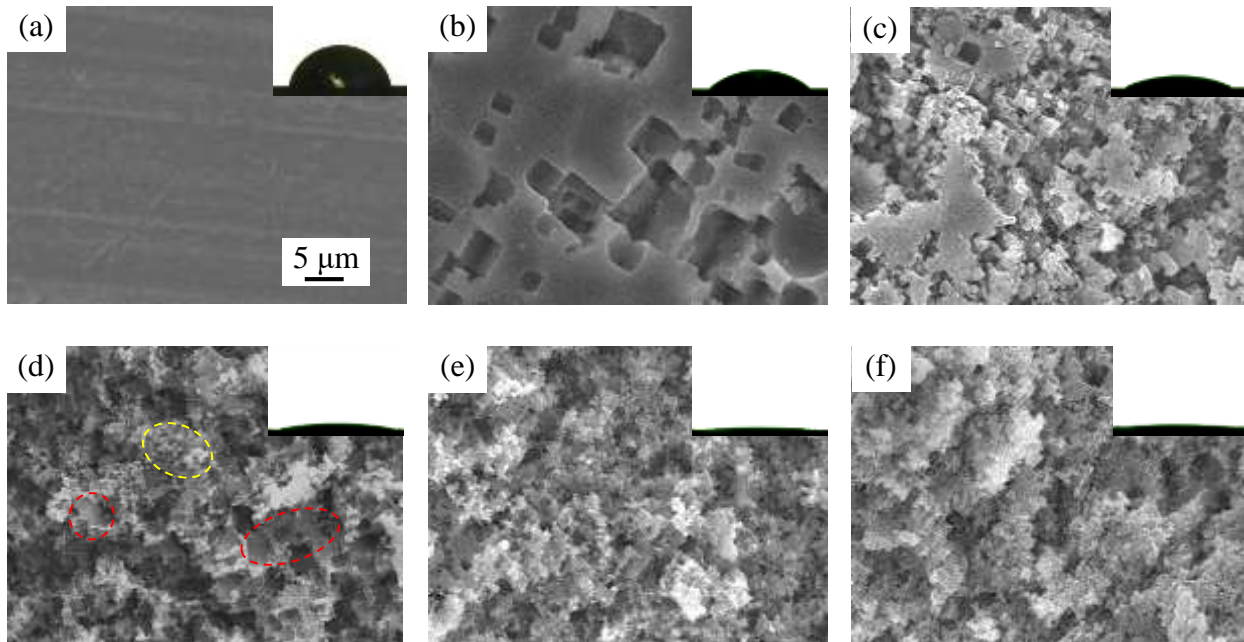
1 The surface morphology was obtained using the scanning electron microscope (SEM, S-3500N,
2 Hitachi, Japan) and confocal laser scanning microscope (CLSM, LEXT OLS 3000, Olympus,
3 Japan). Liquid droplets (3.5 wt% NaCl solution) of 5 μL were gently deposited on the sample
4 surface with a syringe and the side view images of the droplets were obtained with a digital camera
5 (Figure S2). The average static WCA was determined at six different spots on two sides of each
6 sample at room temperature.

7 **Electrochemical Measurements.** The electrochemical properties of the samples were
8 investigated by EIS using an electrochemical workstation (CHI660E, CH Instruments, China). The
9 electrochemical measurements were performed in a three-electrode system comprising a saturated
10 calomel electrode as the reference electrode, a platinum wire as the counter electrode, an Al sample
11 with an exposed surface area of 3.9 cm^2 (15 mm \times 13 mm \times 2 sides) as the working electrode, and
12 a 3.5 wt% NaCl solution as the electrolyte. Prior to the electrochemical measurements, the sample
13 was immersed in the electrolyte for 60 min to obtain a stable open circuit potential (OCP). A
14 sinusoidal perturbation signal of 5 mV amplitude was then applied, and the impedance spectra
15 were recorded at OCP in the frequency range 0.1 Hz–100 kHz using a logarithmic sweep. The
16 electrochemical data were analyzed with the aid of CHI660E, Zsimpwin, and Zview (Scribner
17 Associates Inc.) software.

18 RESULTS AND DISCUSSION

19 **Surface Morphology and Wettability.** Figure 1 shows SEM images of the fabricated samples
20 (a)–(f). Polishing traces can be observed on the bare Al surface, as shown in Figure 1a, and porous
21 layers containing nano/microstructures are observed on the chemically etched Al surfaces, as
22 shown in Figures 1b–1f. It seems that increasing the chemical etching time results in denser

1 nano/microstructures. After 1 s immersion, several microscale square cavities are generated on the
2 surface (Figure 1b). When etching time was extended to 2 s and above, the surface morphology
3 changed drastically and the upper layer of surface is found to have damaged. As etching time
4 prolonged to 5 s, geometrical microconvex plate structures are formed on the surface shown in
5 Figure 1d. Furthermore, denser smaller submicro/nano structures start to generated on the top of
6 microconvex plate (yellow dash circle in Figure 1d) but parts of the surfaces are not etched
7 completely (red dash circle in Figure 1d). By extending the etching time to 10 and 20 s, the
8 generated microconvex plate structures were further etched and hierarchically structures were
9 formed on the surfaces shown in Figures 1e and 1f.



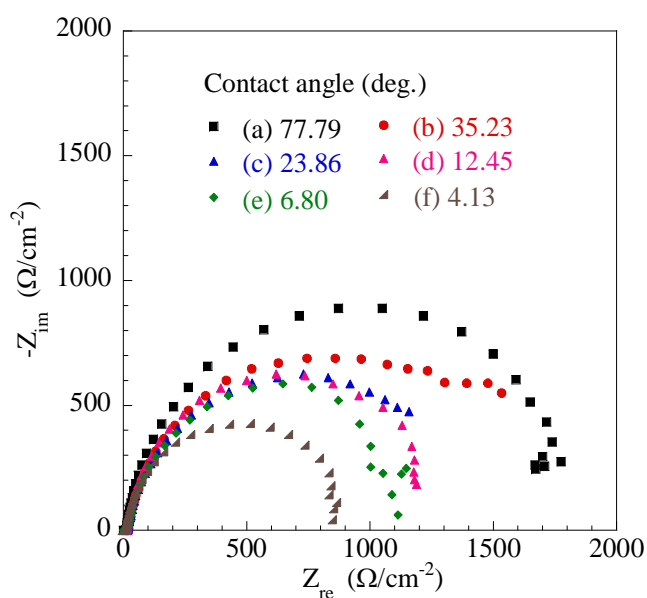
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11 **Figure 1.** SEM images of Al surfaces after immersed in a CuCl_2 solution for (a) 0, (b) 1, (c) 2, (d)
12 5, (e) 10, and (f) 20 s. The insets show the side views of water droplets on the structured surfaces.

13 Side views of droplets are also shown in Figure 1. The WCA for a bare Al surface is $77.29^\circ \pm$
14 2.06° , and surface hydrophilicity is enhanced with increasing chemical etching time. The WCA is

1 $35.23^\circ \pm 0.65^\circ$ for sample (b), $23.86^\circ \pm 1.06^\circ$ for sample (c), $12.45^\circ \pm 1.60^\circ$ for sample (d), 6.80
2 $\pm 1.29^\circ$ for sample (e) and $4.13 \pm 0.79^\circ$ for sample (f). The static contact angle decreases with
3 increasing chemical-etching time, while the contact radius increases in an opposite manner.
4 Furthermore, arithmetic surface roughness (Ra) were assessed from the cross-sectional profiles
5 (Figure S3). The roughness of the surfaces, which is related to wettability, increases with etching
6 time. The average values of Ra for samples (a)–(f) are 0.254 ± 0.039 , 0.809 ± 0.090 , 1.002 ± 0.112 ,
7 1.942 ± 0.053 , 1.969 ± 0.034 , and $2.230 \pm 0.017 \mu\text{m}$, respectively, indicating that denser
8 nano/microstructures are formed with longer etching time.

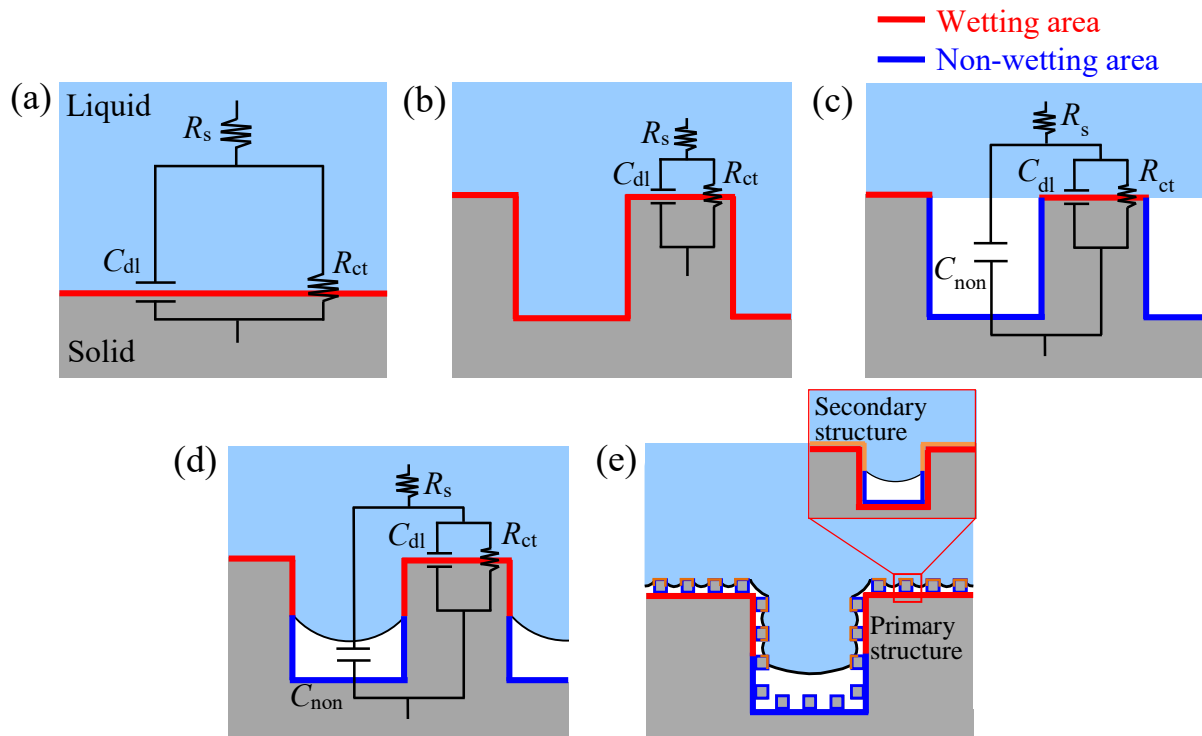
9 **Electrochemical Analysis.** EIS was used to analyze the interfacial electrical impedance of the
10 samples. Typical Nyquist plots, which represent impedance as a complex number with its real part
11 in the x-axis and imaginary part in the y-axis, are shown in Figure 2. The impedance of the bare
12 Al sample with a WCA of 77.79° is the highest, whereas that of the sample etched for 20 s, which
13 has a WCA of 4.13° , is the lowest. That is, electrochemical impedance is correlated with
14 wettability and decreases as WCA decreases.



15

1 **Figure 2.** Nyquist plots for Al surfaces with the contact angle of 77.79°, 35.23°, 23.86°, 12.45°,
 2 6.80°, and 4.13° in 3.5 wt% NaCl solution.

3 Since the defects or inhomogeneities of the electrode surface cause the capacitance components
 4 deviate from ideal (smooth) pure capacitance, the Nyquist plots shown in Figure 2 change from
 5 semi-circles to semi-ellipses. In order to clarify the effect of effective wetting area on the
 6 electrochemical impedance, an EEC with one time constant shown in Figure 3a and 3b was applied
 7 to analyze the measured EIS data of the flat and fully wetted structured surfaces. Here R_s is the
 8 solution resistance, which is dominated by the electrolyte at a given working electrode surface; C_{dl}
 9 is the double-layer capacitance related to the surface structure depending on frequency; and R_{ct} is
 10 the charge transfer resistance determined by the surface conductivity and the polarization
 11 resistance.⁴⁴



12

1 **Figure 3.** Schematic views and equivalent electrical circuits (EECs) for the solid–liquid interfaces
2 of different samples immersed in NaCl solutions: (a) flat surface, (b) fully wetted structured
3 surface, (c) non-wetted structured surface, (d) partial wetted structured surface and (e) partial
4 wetted hierarchically structured surface.

5 Compared to the EECs shown in Figures 3a and 3b, a capacitance is added to the non-wetting
6 area, C_{non} , as a parallel element for the non-wetted and partial wetted structured surface. Therefore,
7 the total equivalent impedance is derived from a parallel combination of the wetted area and the
8 non-wetted area impedances. Figures 3c and 3d show the schematics of solid–liquid interface and
9 the corresponding EECs of non-wetted and partial wetted one-scale structured surfaces, and Figure
10 3e shows that of the hierarchically structured surface. Remarkably, under normal temperature and
11 pressure, the electronic and ionic conductivity of non-wetting area (air or vacuum) is almost 0,
12 which is much smaller than that of electrolyte (NaCl solution). Thus, the electrons and ions cannot
13 easily pass through the non-wetting area, resulting in the impedance modulus of non-wetting area
14 is much larger than that of wetting area ($|Z_{\text{wetting}}| \ll |Z_{\text{non-wetting}}|$). As the total equivalent
15 impedance dependent on the lower impedance at the parallel circuits, which indicate that the solid–
16 liquid contact area plays dominate role in the impedance measurement. As a result, the EECs of
17 the non-wetted and partial wetted structured surfaces can be simplified to that shown in Figure 3a.

18 Considering the defects or inhomogeneities of the samples, a constant phase element (CPE) is
19 used to fit the non-ideal capacitive behavior instead of the electrical capacitance C_{dl} , for the one
20 time constant EEC. The impedance of the CPE (Z_{CPE}) is

$$21 \quad Z_{\text{CPE}} = \frac{1}{Q(j\omega)^n} \quad (3)$$

1 where j is a unit of the imaginary part ($j = \sqrt{-1}$), ω is the angular frequency, $\omega = 2\pi f$ (rad/s), n is
 2 an exponential coefficient, and Q is a constant.^{45,46} Therefore, the impedance Z becomes

$$3 \quad Z = R_s + \frac{R_{ct}}{R_{ct}Q(j\omega)^n + 1} \quad (4)$$

Table 1. Electrochemical data for samples in a 3.5 wt% NaCl solution.

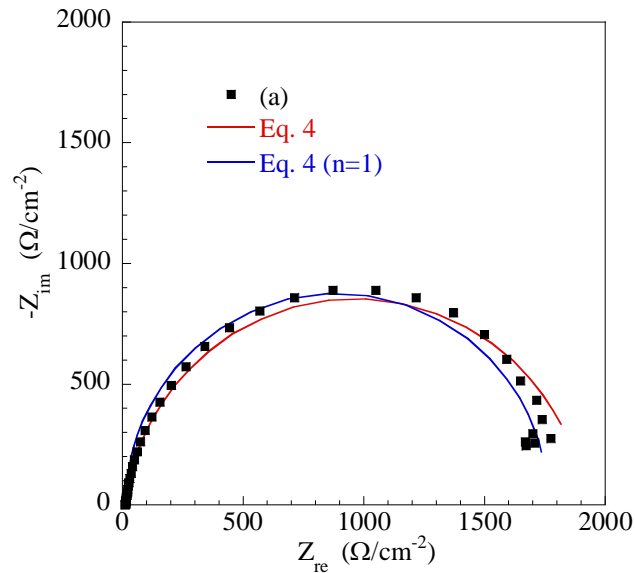
Samples	R_s ($\Omega \cdot \text{cm}^2$)	CPE		R_{ct} ($\Omega \cdot \text{cm}^2$)
		Q ($\Omega^{-1} \text{ s}^n/\text{cm}^2$)	n	
a	12.5	1.517×10^{-5}	0.930	1905.3
b	15.5	1.201×10^{-4}	0.887	1753.0
c	17.4	6.884×10^{-5}	0.900	1461.5
d	13.9	3.903×10^{-5}	0.919	1385.7
e	15.0	3.888×10^{-5}	0.861	1242.5
f	11.5	4.872×10^{-5}	0.913	960.3

4

5 The comparison of Nyquist plots simulated by CPE and that of C_{dl} for bare Al are shown in
 6 Figure 4. The CPE (red line) and C_{dl} plots (blue line) agree well with the experimental results, and
 7 no significant difference was observed. Table 1 summarizes the data analyzed for each sample.
 8 The value of n can be approximated to 1 with a range of 0.861–0.930. Also, the simulated results

1 of CPE and C_{dl} for the etched samples show tendencies similar to those for the bare Al, which is
2 in agreement with the experimental results (Figure S4). Thus, the CPE shows capacitor behavior
3 and can be replaced by C_{dl} .

4 As the impedance spectra were recorded in the frequency range of 0.1 Hz–100 kHz, the
5 influence of frequency on impedance was investigated. At low frequency region, since the
6 wavelength of the disturb signal is long, the ions, electrons are difficult to penetrate the solid–
7 vapor–liquid interface (blue parts in Figure S5a). The charge only transfers at the solid–liquid
8 contact area. On the other hand, the wavelength is short at high frequency region (Figure S5b),
9 ions and electrons are easy to obtain energy, which result in large penetration force. Thus, there is
10 no significant deviation of charge transfer between the solid–liquid interface and solid–vapor–
11 liquid interface. As a result, the impedance at low frequency region was studied to better clarify
12 the effective wetting area at the structured surfaces.



13

14 **Figure 4.** Nyquist plots of the experimental data and those derived from Eq. (4) for the sample (a).

1 Based on the EEC shown in Figure 3a, the impedance Z and the modulus of impedance $|Z|$ are
 2 given by Eqs. (5) and (6), respectively,

$$3 \quad Z = R_s + \frac{R_{ct}}{1 + j\omega R_{ct} C_{dl}} \quad (5)$$

$$4 \quad |Z| = \sqrt{R_s^2 + \frac{R_{ct}^2 + 2R_{ct}R_s}{1 + (R_{ct}\omega C_{dl})^2}} \quad (6)$$

5 The term $(R_{ct}\omega C_{dl})^2$ is small enough to be neglected at low frequency.⁴⁷ Thus, $|Z|$ is

$$6 \quad |Z| = R_s + R_{ct}, \quad (7)$$

7 where $|Z| \approx R_{ct}$ if $R_s \ll R_{ct}$. Defining Z_0 as the impedance of flat bare Al surface (apparent area (A_0)
 8 = 3.9 cm²), the real contact area can be obtained as follows:

$$9 \quad A = \frac{|Z_0|}{|Z|} A_0 \quad (8)$$

Table 2. Impedance elements at low frequency and real contact area.

Samples	$ Z $ ($\Omega \cdot \text{cm}^2$)	R_{ct} ($\Omega \cdot \text{cm}^2$)	$ Z_{c_{dl}} $ ($\Omega \cdot \text{cm}^2$)	R_s ($\Omega \cdot \text{cm}^2$)	A/A_0 (-)	A (cm ²)
a	1917.8	1905.3	1.104×10^4	12.5	1.00	3.90
b	1768.5	1753.0	1.325×10^3	15.5	1.08	4.21
c	1478.9	1461.5	2.331×10^3	17.4	1.30	5.07

d	1399.5	1385.7	4.078×10^3	13.9	1.37	5.34
e	1257.6	1242.5	4.093×10^3	15.0	1.53	5.97
f	971.9	960.3	3.267×10^3	11.5	1.97	7.68

1

2 Table 2 shows the analysis results based on EIS data for the low frequency of 0.1 Hz. The

3 impedance modulus $|Z|$ is almost equal to R_{ct} , and R_s is small enough to be ignored. The impedance

4 of C_{dl} is large at low frequency; however, it can be ignored because it is a parallel element in the

5 EECs. We confirm that R_{ct} is the dominant factor for $|Z|$ at low frequency. Furthermore, the real

6 contact areas for the partial wetting model were estimated and, as shown in Table 2, the effective

7 wetting area increases with decreasing contact angle. The ratio of the real contact area to the

8 apparent area for the sample etched for 20 s is close to 1.97 owing to the denser

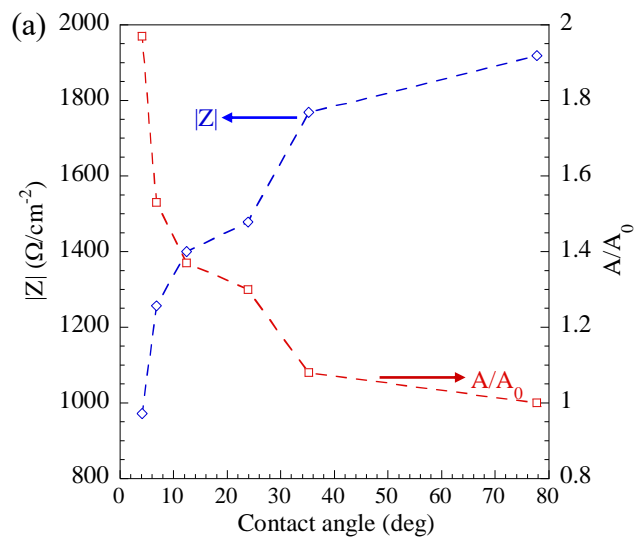
9 nano/microstructures, as shown in Figure 1f. As shown in Figure 5, the total impedance modulus

10 and R_{ct} show a similar correlation with the WCA at low frequency, i.e., a surface with a larger

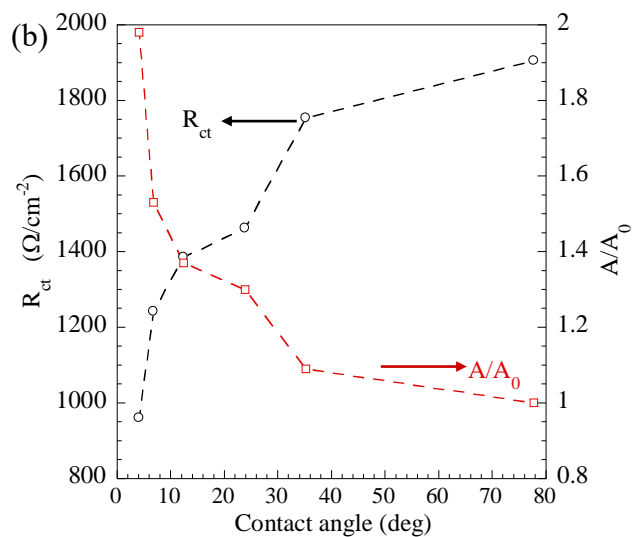
11 WCA has a larger impedance and smaller effective wetting area. Moreover, compared to other

12 elements shown in EECs, R_{ct} is found to be the dominant parameter at low frequency and can be

13 used to estimate the effective wetting area.



1



2

3 **Figure 5.** Effect of contact angle on electrochemical property and real contact area: (a) impedance
 4 modulus and (b) charge transfer resistance (the dotted lines are eye guides).

5 **CONCLUSION**

6 The effective wetting area of the solid–liquid interface at the nano/microstructured surface was
 7 investigated on the basis of the electrochemical impedance analysis in this study. We first applied
 8 the theoretical wetting model to the equivalent electrical circuits for the electrochemical

1 impedance analysis to estimate the effective wetting area of hydrophilic nano/microstructured
2 surfaces. The water contact angle decreases significantly at hydrophilic surfaces with denser
3 nano/microstructures. The electrochemical impedance decreases with reducing water contact angle,
4 showing a dependence on the effective wetting area at the solid–liquid interface. Also, the charge
5 transfer resistance at a low frequency is the most important factor to estimate the effective wetting
6 area at the solid–liquid interface. Further experiments and electrochemical analysis on the effective
7 wetting area at the hydrophobic nano/microstructured surfaces are now under investigation.

8 DISCLOSURE

9 The authors declare no competing financial interest.

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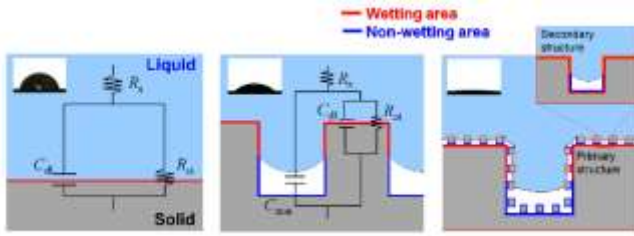
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1 Graphic Abstract



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